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Controllable synthesis of Cu_2O decorated WO_3 nanosheets with dominant (0 0 1) facets for photocatalytic CO_2 reduction under visible-light irradiation



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ABSTRACT

Systematical design and controllable assembly of nanostructured photocatalysts have received much attention in the field of CO $_2$ reduction. The Cu $_2$ O decroted hexagonal WO $_3$ nanosheets with and without dominant (0 0 1) facets (Cu $_2$ O/WO $_3$ -001 and Cu $_2$ O/WO $_3$) were synthesized vertically on the surface of fluorine-doped stannic oxide (FTO) substrate and their photocatalytic performance for CO $_2$ reduction were evaluated in the presence of H $_2$ O vapour under visible light irradiation (λ > 400 nm). The Cu $_2$ O/WO $_3$ -001 catalyst exhibited higher photocatalytic activity than those of Cu $_2$ O, WO $_3$ -001 and Cu $_2$ O/WO $_3$ -01 catalyst exhibited higher photocatalytic activity than those of Cu $_2$ O, WO $_3$ -01 and Cu $_2$ O/WO $_3$ -1 he maximal product yields of CO, O $_2$ and H $_2$ for Cu $_2$ O/WO $_3$ -001 after 24h illumination reached 11.7, 5.7 and 0.7 µmol, respectively, and good cycling ability was discovered after 4 cycles. The (0 0 1) facet of hexagonal phase WO $_3$ nanosheet was in favor of the H $_2$ O oxidation in the CO $_2$ reduction process. Additionally, the Z-scheme charge transfer mode of Cu $_2$ O/WO $_3$ heterojunction could promote photoinduced charge separation and enhance redox ability of the separated electrons and holes, leading to excellent photocatalytic CO $_2$ reduction performance. The study may provide some insights into the coherent design of specific nanosheet photocatalysts with Z-scheme charge transfer for CO $_2$ reduction.

1. Introduction

Photocatalytic reduction of CO_2 to produce valuable chemicals using solar energy has recently received considerable interests due to its great potential for reducing atmospheric CO_2 and supplying renewable fuels [1–3]. Photocatalytic CO_2 reduction requires an efficient catalyst that possesses excellent photo-absorption property in the visible light region and efficient extraction ability of the photogenerated electronhole pairs to reactive sites [4]. However, the efficiency of photocatalytic CO_2 reduction over a single photocatalyst is very low and far from practical application, probably due to the shortages of its inherent physicochemical properties [5]. On going attempts for the construction of an efficient photocatalyst include nanostructuralization [6,7], tailoring of crystal phases [8], construction of heterojunctions [9,10], band-structure engineering [3,11] and so on, and significant progresses have been achieved to date.

Controlled assembly of nanostructured semiconductors offers a promising avenue for developing highly efficient photocatalysts for CO₂ reduction. Particularly, semiconductors with two-dimensional nanosheet structures could lead to short carrier diffusion length and efficient charge separation and migration [6,7], and many photons could be absorbed by the nanosheet in a remarkably short time due to its large surface area, which are favorable for enhancing the photocatalytic activity of multi-electron reaction [12]. It also has been verified that the size-quantization eff ;ects in nanosheets could change the band gap as well as the redox power of the semiconductor as compared with bulk catalysts, and thus modify the catalytic activity and selectivity of surface reaction, especially multi-electron reaction such as CO2 reduction [13]. Furthermore, the construction of heterojunctions, especially typical all-solid-state Z-scheme systems, could promote photoinduced charge separation and enhance redox capacity of photoinduced electrons and holes, and thus improve the photocatalytic activity for CO2

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reduction [10,14,15]. Therefore, it is a practical way to explore effective photocatalysts by the construction of nanosheet-structured semiconductors and Z-scheme heterojunction systems with well-organized alignments.

WO₃ has attracted a great deal of interests in photocatalysis due to the medium band gap ($E_g = 2.4 \sim 2.8 \text{ eV}$), high oxidizing power of the valence band holes and good stability in acidic and oxidative conditions [16]. In particular, WO3 nanosheets could be more effective in promoting charge separation and thus resulting in enhanced photocatalytic activity [17,18]. Chen et al. [11] reported that monoclinic WO3 nanosheet exhibited better photocatalytic activity for CO2 reduction than that of commercial WO₃ powder. Monoclinic and hexagonal WO₃ are the most attractive candidates among the various crystal phases. The monoclinic phase is the most stable phase, while the hexagonal phase possesses open-tunnel structure and rich intercalation chemistry [19]. Meanwhile, different crystalline structures of WO3 could result in dissimilar electronic properties, and thus cause differences in photocatalytic activity for CO2 reduction [20,21]. For example, Ohno et al. [22] showed that the monoclinic WO₃/g-C₃N₄ with Z-scheme exhibited good photocatalytic activity due to the facilitated transfer of electrons from CO₂ to CH₃OH. Ying et al. [23] has proved that the (0 0 2) facets of monoclinic WO3 could play a positive effect on the photocatalytic activity for CO₂ reduction. Furthermore, it is believed that the (0 0 1) facets of hexagonal WO3 with suitable channels could be filled with H₂O molecules, and act as fantastic chains for facilitating the electron transfer [4], which is theoretically favorable for photocatalytic reduction of CO2 with H2O vapour [3]. However, the flat-band potential of electrons in the conduction band of hexagonal WO3 is lower than that required for CO2 reduction, resulting in the inability to reduce CO2 solely by hexagonal WO3 [18]. Hence, single hexagonal WO3 photocatalyst has rarely been reported for CO2 reduction. To solve these concerns, heterogeneous photocatalysts could be developed to further modify the redox capacity and photoinduced charge separation of the hexagonal WO₃-based composites. Cu₂O, a p-type semiconductor with a band gap of about 2.1 eV, is an attractive photocatalyst for CO2 reduction [24-26], due to its intensive visible-light absorption ability and appropriate conduction band position (about -1.1 V vs. NHE). [11,20] Therefore, an effective way to address these issues may be provided by constructing a Cu₂O/WO₃ Z-scheme system for photocatalytic CO₂ reduction.

To the best of our knowledge, few studies have been done over hexagonal WO $_3$ -based catalysts for CO $_2$ reduction. Herein, in this research, hexagonal WO $_3$ nanosheets with and without dominant (0 0 1) facets (WO $_3$ -001 and WO $_3$) were synthesized by the seed-mediated hydrothermal methods, and Cu $_2$ O was electrodeposited onto the WO $_3$ nanosheets to fabricate Cu $_2$ O/WO $_3$ -001 and Cu $_2$ O/WO $_3$ composite photocatalysts. Photocatalytic activities for CO $_2$ reduction with H $_2$ O vapour were investigated, and the charge transfer mechanism for photocatalytic CO $_2$ reduction was further proposed for the WO $_3$ -based catalysts.

2. Experimental section

2.1. Synthesis of hexagonal WO_3 nanosheets with and without dominant (0 0 1) facets

All chemicals used in the experiments were of analytical grade and used without further purification. Hexagonal WO $_3$ nanosheets with and without dominant (0 0 1) facets were both synthesized on the fluorine-doped tin oxide (FTO) glass substrates through seed-mediated hydrothermal methods. [27] A H $_2$ WO $_4$ solution for the seed layer and hydrothermal growth was prepared by dissolving 1.25 g of H $_2$ WO $_4$ into 30 wt% H $_2$ O $_2$ and heated at 95 $^{\circ}$ C for 2 h. Before the hydrothermal growth, a seed layer was deposited onto the FTO substrate by spinning the above H $_2$ WO $_4$ solution with subsequent annealing treatment at 500 $^{\circ}$ C in air for 2 h. In a typical synthesis of WO $_3$ nanosheets with dominant (0

0 1) facets, a seeded FTO substrate was put into a Teflon-lined stainless steel autoclave with vertically orientation. Then, a mixed solution containing $3.0\,\mathrm{mL}$ of $\mathrm{H_2WO_4}$ solution, $0.5\,\mathrm{mL}$ of HCl (6 M) and $12.5\,\mathrm{mL}$ of acetonitrile was added into the reactor for the hydrothermal reaction at $180\,^{\circ}\mathrm{C}$ for 6 h. Afterward, the as-prepared sample was rinsed with deionized water, dried in air, and annealed at $500\,^{\circ}\mathrm{C}$ for $0.5\,\mathrm{h}$. The obtained sample was denoted as $\mathrm{WO_3}\text{-}001$. Hexagonal phase $\mathrm{WO_3}$ nanosheets without dominant (0 0 1) facets were prepared using the similar procedure, except that the hydrothermal solution was prepared by mixing $3.0\,\mathrm{mL}$ of $\mathrm{H_2WO_4}$ solution, $1.5\,\mathrm{mL}$ of HCl (5 M) and $10.5\,\mathrm{mL}$ of acetonitrile together, and the obtained sample was denoted as $\mathrm{WO_3}$.

2.2. Synthesis of Cu₂O/WO₃ samples

 Cu_2O was electrodeposited on the surface of WO_3 -001 and WO_3 nanosheets using a modified method. [28] Typically, the deposition was conducted at -0.4 V in 0.5 M $Cu(OAc)_2$ aqueous solution using a standard three-electrode system, which was composed of WO_3 -001 and WO_3 nanosheet working electrodes, a platinum counter electrode, and a Ag/AgCl reference electrode. After deposition, the obtained samples were rinsed with deionized water and dried in air. The as-prepared materials were denoted as Cu_2O/WO_3 and Cu_2O/WO_3 -001. The control samples Cu_2O/WO_3 -001-m and Cu_2O/r - WO_3 -001 were also synthesized and the detailed process was described in the supplementary material.

2.3. Catalysts characterization

The crystal phases of the samples were recorded with X-ray diffractometer (PANalytical X' pert PRO, Netherlands). The BET surface areas were determined by N2 adsorption-desorption isotherm measurements at 77 K (Autosorb-iQ, USA), and low-pressure CO2 adsorption isotherms were conducted at 273 K for both WO3 and WO3-001. The pore-size analysis of the prepared sample was based on a nonlocal density functional theory (NLDFT) model. A scanning electron microscope (SEM, Nova NanoSEM 450, FEI) was used to characterize the morphologies of the obtained samples. Transmission electron microscopy (TEM) images were obtained on a Tecnai G² F20 S-TWIN electron microscope. Furthermore, high-resolution transmission electron microscopy (HRTEM) and energy dispersive X-Ray spectroscopy (EDX) were employed. X-ray photoelectron spectroscopy (XPS) measurements were carried out at room temperature on a Thermo escalab 250Xi X-ray Photoelectron Spectrometer. UV-vis absorption spectra were performed on a Cary 5000 scan UV-vis spectrometer.

2.4. Photocatalytic performance tests

The photocatalytic experiment for the reduction of CO2 with H2O vapour was carried out in a stainless steel cylindrical vessel with a quartz window on top of the reactor. A 300 W Xenon arc lamp (PLS-HXF 300, Beijing China Education Au-light Co. Ltd., China) with a UV cutoff filter ($\lambda > 400$ nm) was utilized as the light source. The catalyst (about 85 mg was placed on a glass sheet $(1.8 \times 2.0 \text{ cm}^2)$ inside the reactor. Prior to light irradiation, the reaction setup was vacuum treated, and then purged with high purity CO2 gas (99.995%) for several times. The compressed high purity CO2 gas was passed through a water bubbler to generate a mixture of CO2 and H2O vapour. After illumination, the gaseous products were quantifiably identified using an off-line GC-7890II gas chromatograph (Techcomp Corp., China) equipped with both a flame ionization detector (FID) and a thermal conductivity detector (TCD). The equipped columns were TDX-01 and Porapak-Q, respectively. Then, other hydrocarbon products were analyzed by GC-MS (Agilent 7890A-5975C) equipped with a DB-FFAP capillary column. The isotope-labeled experiments were performed using ¹³CO₂ instead of ¹²CO₂, and the products were analyzed using GC-MS. In order to evaluate the stability, the composites were refreshed by the electrochemical treatment before washing and drying,

and their photocatalytic performances were reassessed. Photoelectrochemical test was conducted in a conventional three-electrode system using a CHI 660E electrochemical workstation (Chenhua, Shanghai, China). Photocurrents of the photocatalysts were measured at 0.0 V (ν s. Ag/AgCl) in 0.5 M Na₂SO₄ aqueous solution under light illumination, and dissolved oxygen was removed by purging with high purity nitrogen gas before the measurements. Apparent quantum efficiency (AQY) of the as-prepared samples at 420 nm was defined by the following equation because of a 2-hole process for the production of CO as a main product from CO₂ reduction: [22]

$$AQY(\%) = \frac{2 \times N_{CO}}{Number\ of\ incident\ photons} \times 100\%$$
(1)

* N_{CO} represented the number of evolved CO molecules.

In the photocatalytic O_2 evolution experiment, $80\,\mathrm{mg}$ of photocatalyst was suspended in $100\,\mathrm{mL}$ of water with and without adding KBrO $_3$ electron scavenger. The reactant solution was thoroughly degassed under flowing high-purity N_2 for $90\,\mathrm{min}$ in prior. The reaction solution was magnetically stirred and the temperature of the reaction system was maintained below $4\,^\circ\mathrm{C}$ during the entire experiment. The evolved O_2 was analyzed through a gas chromatographer (Agilent GC-6890, $5\,\mathrm{A}$ molecular sieve column) equipped with a thermal conductivity detector using high-purity Ar (99.999 vol%) as the carrier gas.

3. Results and discussion

3.1. Composition, structure and morphology

XRD patterns of the as-prepared materials were shown in Fig. 1. WO₃ sample exhibited pronounced diffraction peaks at 13.9°, 22.7°, 24.3°, 26.8° and 28.2°, corresponding to (1 0 0), (0 0 1), (1 1 0), (1 0 1) and (2 0 0) facets of hexagonal phase WO₃ (JCPDS No. 33-1387). The peak intensity of (0 0 1) facets for WO₃-001 nanosheet was obviously enhanced, which was caused by exposing the specific fecats. By measuring the respective peak areas of the (1 0 0), (0 0 1) and (2 0 0) facets, the fraction of (0 0 1) facets from XRD patterns was estimated by the following equation (Eq. (1)). [27]

$$w_{(0\ 0\ 1)} = \sum A_{(0\ 0\ 1)} / \left[\sum A_{(1\ 0\ 0)} + \sum A_{(0\ 0\ 1)} + \sum A_{(2\ 0\ 0)}\right]$$
(2)

The calculated percentage of (0 0 1) facets for WO $_3$ and WO $_3$ -001 are about 25% and 68%, respectively. Additionally, cubic Cu $_2$ O (JCPDS No. 01-078-2076) was successfully electrodeposited onto the WO $_3$ -001 and WO $_3$ samples, respectively. The obtained Cu $_2$ O/WO $_3$ -001 and Cu $_2$ O/WO $_3$ composites exhibited characteristic diff ;raction peaks of both WO $_3$ and Cu $_2$ O, and no impurity peaks were detected, indicating that the deposition of Cu $_2$ O didn't alter the crystalline phase of WO $_3$.

X-ray photoelectron spectroscopy (XPS) measurements were carried

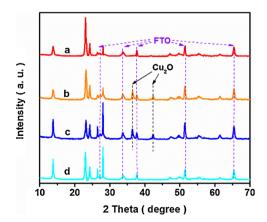


Fig. 1. XRD patterns of WO $_3$ -001 (a), Cu $_2$ O/WO $_3$ -001 (b), Cu $_2$ O/WO $_3$ (c) and WO $_3$ (d).

out to elucidate the surface composition and chemical states of the samples. The survey XPS spectra (Fig. S1) indicated that the composites were composed of W, Cu and O elements. The high resolution XPS spectra of W 4f and Cu 2p in Fig. 2 were deconvoluted by the Gaussian-Lorenzian method. As shown in Fig. 2a, two distinct peaks centered at binding energies of 35.5 and 37.7 eV corresponded to W⁶⁺ oxidation state of WO₃-001, while the relatively weak peaks at 34.5 and 36.5 eV were attributed to W⁵⁺ oxidation state, [29,30] which was necessary to maintain its hexagonal phase structure [31]. Besides, Cu₂O/WO₃-001 exhibited higher W⁵⁺/W⁶⁺ ratio than that of WO₃-001 from Fig. 2a, which indicated that W⁶⁺ species were partially reduced during the electrodeposition of Cu₂O. As shown in the high resolution Cu 2p XPS spectra (Fig. 2b), the Cu₂O/WO₃-001 sample exhibited two characteristic peaks centered at 932.6 and 952.3 eV, corresponding to Cu⁺ 2p_{3/2} and $2p_{1/2}$, respectively. Two weak peaks were also observed at binding energies of 933.5 eV for $\text{Cu}^{2+}\ 2p_{3/2}$ and 953.6 eV for $\text{Cu}^{2+}\ 2p_{1/2}$ [32], and the other two tiny shakeup satellite peaks centered at 940.4 and 960.7 eV further confirmed the coexistence of a trace amount of CuO having a d⁹ configuration in the ground state, which may be due to surface oxidization of Cu₂O in air. [33] Furthermore, the Cu 2p and W 4f peaks of Cu₂O/WO₃-001 were positively shifted in comparison with those of pure Cu₂O and WO₃-001, which can be attributed to the intense interaction between WO₃-001 and Cu₂O, implying that the heterojunctions have been formed in the as-synthesized composites.

The morphologies and micro-structural information of the as-prepared samples were visualized by SEM and TEM measurements. Fig. 3 showed SEM and TEM images of the as-prepared samples. As can be seen, nanosheets were successfully grown on FTO substrates for WO₃ (Fig. 3a) and WO₃-001 (Fig. 3b), and free spaces were clearly observed among the nanosheets, which were beneficial for full contact between catalysts and reactant gas. The TEM images of WO₃-001 (Fig. 3c) revealed a typical nanosheet morphology, and the HRTEM image (inset of Fig. 3c) of WO₃-001 showed lattice fringes with interplanar spacings of 0.634 and 0.366 nm, which were consistent with (1 0 0) and (1 1 0) facets of hexagonal phase WO3, respectively. As shown in Fig. S2a, the WO₃-001 showed a maximal pore peak at 0.41 nm, which may be due to the pore structure on the (0 0 1) facets of hexagonal WO3. Additionally, the CO₂ adsorption isotherm of WO₃ exhibited a large uptake at low relative pressure (Fig. S2b), which indicated the presence of microsized tunnels in the structure of the hexagonal phase WO3 nanosheet [34], and the CO₂ adsorption ability of WO₃-001 was stronger than that of WO₃ sample. Based on hexagonal phase WO₃, six tungsten oxide octahedral building blocks sharing their corners assemble into one six-membered ring with a diameter of 0.53 nm [35]. The analysis resulted in the plane of WO3 nanosheet along with the [0 0 1] axis direction, which indicated that the (0 0 1) fecat was explored on the nanosheet surface. Furthermore, the Cu₂O/WO₃ (Fig. 3d) and Cu₂O/ WO₃-001 (Fig. 3e) exhibited similar nanosheet morphologies as compared with those of bare WO3 and WO3-001. The pronounced diffraction fringe of the HRTEM image (inset of Fig. 3f) showed d-spacings of 0.246 and 0.213 nm, corresponding to the (1 1 1) and (2 0 0) facets of cubic Cu₂O, respectively [36,37]. To further explore the detailed elemental distributions, energy-filtered W, Cu, and O maps of the Cu₂O/ WO₃-001 sample were constructed and shown in Fig. S3. As can be seen, all the elements were homogenously distributed, indicating that Cu₂O was uniformly deposited on the surface of WO₃ nanosheet by the electroreduction method.

3.2. Photocatalytic performance for CO₂ reduction

The photocatalytic CO_2 reduction activities of the catalysts were evaluated in the presence of H_2O vapour under continuous visible-light irradiation ($\lambda > 400 \, \text{nm}$) for 24 h. As shown in Fig. 4, no products were observed after illumination for 18 h over Cu_2O , WO_3 and WO_3 -001 catalysts. However, the CO, O_2 and H_2 products were detected for the composite catalysts. To provide solid evidence to validate the origin

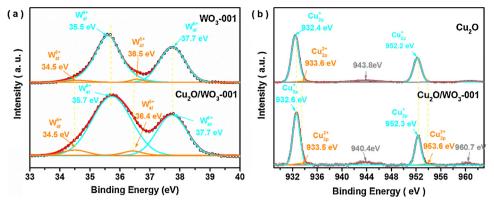


Fig. 2. W 4f (a) and Cu 2p (b) XPS spectra of Cu₂O, WO₃-001 and Cu₂O/WO₃-001.

of the generated CO, a ¹³C labelled isotropic experiment was carried out by using ¹³CO₂ to replace ¹²CO₂. The produced CO was analyzed by GC-MS, and the results were shown in Fig. S4. As can be seen, when $^{13}\text{CO}_2$ was used as the reactant, ^{13}CO (m/z=29) was detected, which was significantly different from the detected ^{12}CO (m/z=28) when using 12CO2 as the reactant. For comparison, the photocatalytic performance of the mechanically dispersed Cu₂O/WO₃-001-m sample was investigated to corroborate the heterojunction effect, and the results showed that tiny amounts of potential products such as CO and O₂ were detected for the mechanically dispersed Cu₂O/WO₃-001-m sample, which indicated that the loosely contacted interfaces were insufficient for photocatalytic reduction of CO2. The results of this isotropic experiment validated that the composite catalysts indeed effectively accelerate photocatalytic transformation of CO₂ to CO. Besides, the XPS spectra of the photocatalysts after illumination were deconvoluted, and the peak area ratios of Cu^{2+}/Cu^+ and W^{5+}/W^{6+} have been summarized in Table S1. The W^{5+} content in composites distinctly decreased, compared with that in pure WO₃-001. It indicated the possible formation of heterojunction between Cu₂O and WO₃-001 for the composite. Moreover, Cu₂O/WO₃-001 composite exhibited higher photocatalytic activity for CO2 reduction than that of Cu2O/WO3 catalyst and the W5+ content kept a lower value than those in other catalysts, which implied that the dominant (0 0 1) facets in WO₃-001 nanosheet could inhibit the

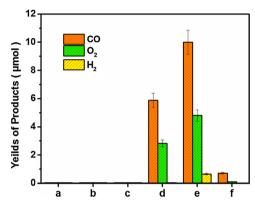


Fig. 4. Photocatalytic performance of CO_2 reduction over Cu_2O (a), WO_3 (b), WO_3 -001 (c), Cu_2O/WO_3 (d), Cu_2O/WO_3 -001 (e) and Cu_2O/WO_3 -001-m (f) samples under continuous visible-light irradiation ($\lambda > 400$ nm) for 18 h.

reduction of W^{5+} in composites. The phenomena may be attributed to the following factors. Firstly, for the Cu_2O/WO_3 -001 sample, the six tungsten oxide (WO_6) octahedral building blocks of hexagonal phase WO_3 shared their corners and assembled into one six-membered ring on (0 0 1) facets [34], and the interstitial spaces could readily

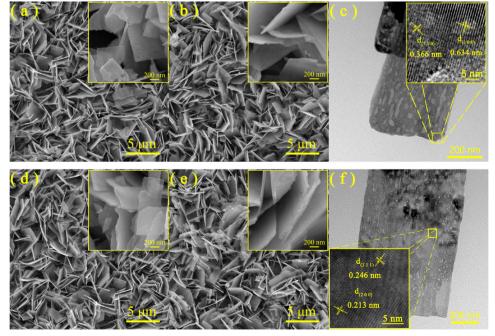


Fig. 3. SEM and TEM images of WO₃ (a), WO₃-001 (b, c), Cu₂O/WO₃ (d) and Cu₂O/WO₃-001 (e, f) samples.

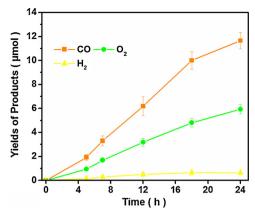


Fig. 5. Total yields of the products over the $\text{Cu}_2\text{O}/\text{WO}_3\text{-}001$ catalyst with different irradiation times.

accommodate guest molecules and ions (e.g. H₂O and H⁺). [35] To investigate the influence of (0 0 1) facet on H2O oxidation, the photocatalytic activity for O₂ evolution was measured and shown in Fig. S5. Due to the weak reduction ability of photoinduced electrons, few O2 and H₂ were produced in pure water for WO₃ and WO₃-001. With the addition of KBrO₃, as an electron scavenger, the photocatalytic O₂ yield of WO₃-001 was distinctly higher than that of WO₃. It indicated that the (0 0 1) facets enabled effective hole conduction throughout the micropore channels of WO₃-001 [35], Besides, the apparent quantum efficiencies (AQY) of the as-prepared composites at 420 nm were summarized in Table S2. The AQY of Cu₂O/WO₃-001 reached 0.503%, which was about twice that of Cu2O/WO3. Hence, the (0 0 1) facet of hexagonal phase WO3 was in favor of the transfer between the photoinduced charges and the adsorbed water molecules, conducing to H₂O oxidation in the CO₂ reduction process, and thus Cu₂O/WO₃-001 exhibited better photocatalytic activity for CO2 reduction.

Fig. 5 showed the photocatalytic activity of the Cu₂O/WO₃-001 sample for CO₂ reduction. As can be seen, CO was detected as the major reduction product, and small amounts of H₂ and O₂ were also generated, which indicated that the photocatalytic reduction of CO₂ was thermodynamically favorable for the formation of CO through a two-electron reduction, due to the proton-assisted multi-electrons pathway of CO₂ reduction [38]. The total yields of CO, H₂ and O₂ enhanced with the prolongation of the irradiation time to 24 h, and finally reached 11.7, 5.7 and 0.7 μ mol, respectively. However, a slight decrease of the production rates occurred throughout the whole process. The maximal production rates of CO and O₂ were found at 18 h to be 0.56 and 0.27 μ mol·h $^{-1}$, respectively (Fig. S6). The recession of the photocatalytic yield rates may be due to oxidation of the products or formation of intermediates covering the surface active sites of photocatalysts.

Recycling photostability of the Cu₂O/WO₃-001 catalyst was investigated, and the results were shown in Fig. 6. As can be seen, the yields of CO, H2 and O2 decreased slightly after four cycles in spite of the electrochemical regeneration. Typically, the yield of CO decreased to 10.1 µmol after the fourth cycle, about 86.3% of that of the first run. XPS analysis was further applied to investigate the stability of the catalyst. In the high resolution XPS spectra of Cu₂O/WO₃-001 catalyst after the fourth cycle (Fig. S7 and Table S3), the peak positions of Cu 2p and W 4f remained nearly unchanged as compared with those of the fresh catalyst. Unfortunately, an obvious raising of W⁵⁺/W⁶⁺ ratio was observed after four cycles. For exploring the role of W5+ for photocatalysis, a large amount of W⁵⁺ was obtained for the controlled sample Cu₂O/r-WO₃-001 by an electro-reduction method. Based on the XPS measurement (Fig. S8), a mass of W⁵⁺ existed in the Cu₂O/r-WO₃-001, and scarcely any products including CO, CH4, H2 and O2 were found, based on the GC–MS results. Hence, the excess $\mathrm{W}^{5\,+}$ in WO_3 component

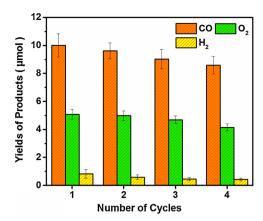


Fig. 6. Recyclability test of the ${\rm Cu_2O/WO_3}$ -001 catalyst under visible-light illumination for 18 h.

led to the decline of photocatalytic performance.

The mechanism of photocatalytic CO₂ reduction was related to the redox abilities of photo-induced electrons and holes, which were influenced by the energy positions of conduction and valence band edges. We determined the electronic band structures of the photocatalysts by UPS and DRS spectra. As shown in Fig. S9, the calculated E_g values for WO₃, WO₃-001 and Cu₂O were 2.75, 2.86 and 2.05 eV, respectively. Fig. 7 showed the UPS spectra of Cu₂O, WO₃ and WO₃-001 samples measured at the same voltage of -8.0 V. According to the linear intersection method, [38] the valence band edge energy (E_{VB}) of Cu_2O was estimated to be -5.56 eV (vs. vacuum), and the conduction band edge energy (E_{CB}) was calculated to be -3.57 eV (vs. vacuum) from the E_{g} and E_{VB} values. The corresponding E_{VB} and E_{CB} values of Cu_2O were 0.91 and -1.12 eV (vs. NHE), respectively, based on the connection between the normal electrode potential (E^{θ}) and the vacuum energy (E_{abs}), E_{abs} = $-E^{\theta}$ - 4.44 [33]. Similarly, the calculated values of E_{CB} and E_{VB} for WO₃-001 were 0.02 and 2.88 eV (vs. NHE) and those of WO₃ were 0.15 and 2.90 eV (vs. NHE). These band energies agreeed well with the previously reported values [39], and there were no distinct differences for the band energies between WO3 and WO3-001. A staggered band alignment heterostructure was successfully formed, assuming that the possible band bending of the semiconductor is neglected.

Transient photocurrent responses were further recorded over several on-off irradiation cycles to demonstrate the separation of photogenerated charge carriers. As shown in Fig. S10, the photocurrents of the catalysts reproducibly increased under irradiation and quickly recovered in the dark. Besides, the photocurrent of the $\text{Cu}_2\text{O}/\text{WO}_3\text{-}001$ sample was about 5 and 15 times higher than those of pure $\text{WO}_3\text{-}001$ and Cu_2O , indicating the reduced recombination and efficient separation of electron-hole pairs in the composite.

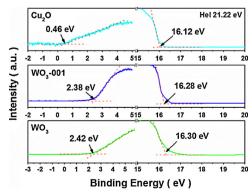


Fig. 7. UPS spectra of the as-prepared Cu₂O, WO₃ and WO₃-001 samples.

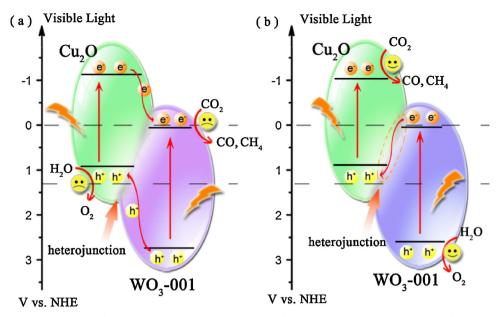


Fig. 8. Schematic illustration of the proposed charge transfer mechanisms: the common charge transfer mode (a) and Z-scheme charge transfer mode (b) for Cu_2O/WO_3 -001.

3.3. Possible photocatalytic mechanism

On the basis of the experimental results and the band energy structure of the Cu₂O/WO₃-001 sample, the enhanced photocatalytic CO₂ activity can be deduced as follows. The p-n junction could be formed at the interface of WO₃-001 and Cu₂O, resulting in the inner electrical field established in the direction from the n-type WO₃ to the p-type Cu₂O. When exposing to the simulated light, the VB electrons of both WO₃-001 and Cu₂O could be excited. The possible separation process of the photoexcited electron-hole pairs at the interfacial phases can be schematically described in Fig. 8a and 8 b. If the photoexcited charge carriers transfered adopting the common double charge transfer mode according to Fig. 8a, the accumulated electrons in the CB of WO₃-001 regrettably cannot realize the CO₂ reduction to CO or H₂O reduction to H2, which was due to that the calculated CB edge potential of WO3 (0.19 V vs. NHE) was more positive than the potential for CO2 reduction (CO/CO₂ = -0.11 V vs. NHE) [3]. Simultaneously, the VB edge potential of Cu₂O (0.91 V vs. NHE) was more negative than that of H_2O oxidation $(O_2/H_2O = 1.23 \text{ V } \text{ vs. NHE})$, and O_2 production cannot be observed. Furthermore, the photo-induced electrons in the CB of WO₃ could cause self-reduction of W⁶⁺ to W⁵⁺ in WO₃ component. According to the promoted photocatalytic activity, the separation of photogenerated carriers was thought to be followed by the Z-scheme mechanism as shown in Fig. 8b. The photo-induced holes tend to remain in the VB of WO_3 , and the electrons in the CB of WO_3 were transferred to the VB of Cu₂O, which results in effective charge separation. H₂O oxidation can be achieved by the photo-induced holes in the VB of WO3, and CO2 with H2O vapour can be reduced by the accumulated electrons in the CB of Cu₂O. Photocatalytic reduction of CO₂ into CO required two electrons, which was dynamically favored than the formation of other fuels such as CH₄ and CH₃OH. Products such as CO, H2 and O2 can be detected for the photocatalytic CO2 reduction with H₂O vapor under visible-light illumination in the experiments. Based on the decline of W⁵⁺/W⁶⁺ ratio in the Cu₂O/WO₃-001 composite as compared with those in the Cu₂O/WO₃ and WO₃-001 samples (Table S1), the W⁵⁺ formation was inhibited in the Cu₂O/WO₃-001 composite. It indicated that the photogenerated electrons in WO₃ component was effectively derived by the heterojunction. Consequently, the Cu₂O/WO₃-001 composite was a direct Z-scheme photocatalyst.

4. Conclusions

In summary, hexagonal-phase WO3 nanosheet with dominant (0 0 1) facets was synthesized by a solvothermal-calcination method, and Cu₂O was composite and the electrodeposition method. The Cu₂O/ WO₃-001 composite nanosheets exhibited an excellent photocatalytic CO_2 reduction performance under visible irradiation ($\lambda > 400 \, \text{nm}$). The yields of CO, O2 and H2 using Cu2O/WO3-001 photocatalysts reached 11.7, 5.7 and 0.7 µmol, respectively, after 24 h visible-light illumination and the photocatalyst showed good photocatalytic activity in the fourth cycling test. The enhanced photocatalytic activity was ascribed to the efficient separation of the photoinduced charges derived from the constructed heterojunction in the nanosheet composite. The (0 0 1) facet of hexagonal phase WO₃ was in favor of the transfer between the photo-induced holes and the water molecules, conducing to H2O oxidation in the CO2 reduction process. The indirect Z-scheme charge transfer mode of Cu₂O/WO₃ was then demonstrated by measuring band structures and the potential mechanism of CO2 reduction is further proposed. The present research is expected to be useful in developing Zscheme photocatalysts of WO3-based nanosheet, and provides meaningful information for hopefully stimulating more insightful investigations of systematical design and controlled assembly of visible-lightdriven photocatalysts.

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Appendix A. Supplementary data

Supplementary material related to this article can be found, in the online version, at doi:https://doi.org/10.1016/j.apcatb.2018.09.076.

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